

# Hierarchical lightweight composite materials for structural applications

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Hierarchical design down to the nanoscale has become possible in structural composite materials with the discovery of carbon nanomaterials such as carbon nanotubes (CNTs) and graphene. Composites that simultaneously combine microscopic continuous fibers and nanoscale reinforcements are known in the field as hierarchical or nanoengineered composites. The additional reinforcement at the nanoscale promises high-performance composites with unique combinations of mechanical properties and new functionalities. Here, we review advances in fiber-reinforced polymers modified with CNTs. Three routes for integration of CNTs in composites are discussed: deposition on fibers/plies, dispersion in the matrix, and assembly into fibers. We highlight opportunities and challenges focusing on mechanical performance and processing.

## Introduction

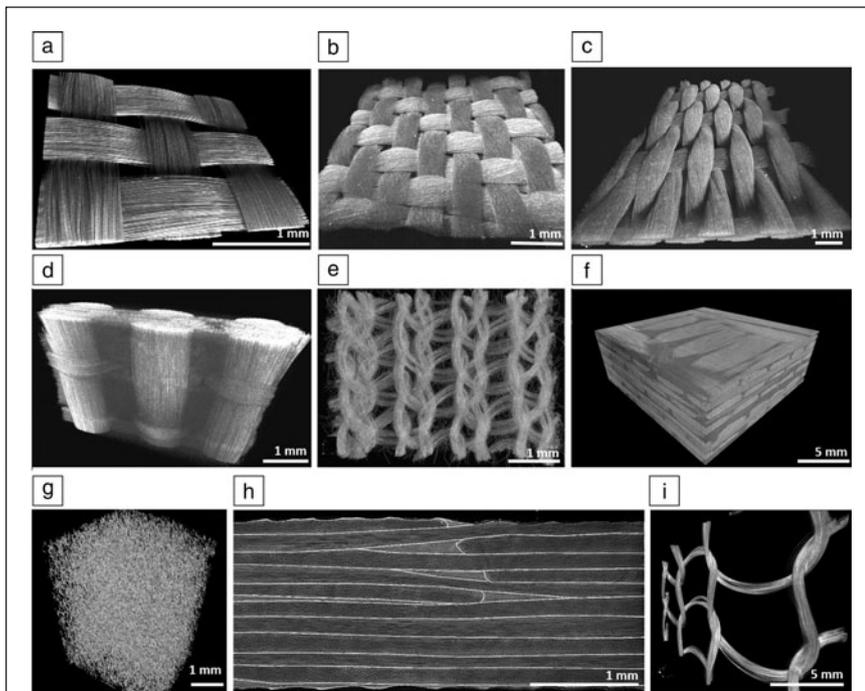
We fly safe airplanes, enjoy the comfort of fast trains, and live in houses powered by wind turbines. These and many other structural applications that employ advanced composite materials are now part of our daily lives. The materials required for these applications must be able to withstand high stresses and serve for decades without failure. Because of increased concerns with air pollution and global warming, additional restrictions on efficiency and environmental impact have to be respected. In the language of engineers, these structural materials must be light, stiff, strong, and durable.

Fiber-reinforced polymer composites and, specifically, carbon fiber-reinforced plastic (CFRP) have been shown to outperform other materials in combining all of these properties. This is evidenced by their significant use in aerospace applications;<sup>1,2</sup> ~50% of the structural mass of the newest airplanes is comprised of such advanced composites. These materials take advantage of the high specific stiffness and strength of microscale continuous fibers, and a compliant polymer matrix, both of which are also lightweight. The typical density of CFRP is about 1.6 g/cm<sup>3</sup>. The microfibers can be layered, woven, braided, knitted, or randomly assembled into a variety of mesolevel morphologies (**Figure 1**) that allow tailoring of composite performance for different applications.

The vision for the next generation of composite materials is that they will be hierarchically designed down to the nanoscale. The importance of nanoscale features has long been recognized from studies of biological materials, which are all truly multiscale composites.<sup>2–4</sup> In addition to using a plethora of different materials, nature also employs structural hierarchy to target multiple properties at the same time, which cannot be otherwise achieved (i.e., both high strength and high toughness). With the discovery of nanomaterials like carbon nanotubes (CNTs) and graphene, it is now possible to take the first steps toward hierarchical designs in man-made materials. Nanoengineering promises structural composites with superior mechanical properties as well as new functionalities. This review focuses on opportunities and challenges for polymer-based hierarchical composites of high stiffness, strength, and toughness enabled by CNTs.

Tubular nano-sized carbon filaments were reported more than six decades ago.<sup>5,6</sup> It was not until Iijima's report on CNTs in 1991<sup>7</sup> that the composites community began intensively exploring CNTs. Interest in CNTs as reinforcement of polymers is due to their high stiffness (1 TPa) and strength (on the order of 50 GPa),<sup>8</sup> both of which are superior to those of existing carbon fibers (~200–800 GPa)—the most advanced fibers available today. Moreover, high failure strain (12%)<sup>8</sup> has been reported for multiwall CNTs, promising structural composites with some ductility.

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**Figure 1.** Diversity of reinforcement architectures on the mesolevel from 3D microcomputed tomography visualizations. (a) Carbon fiber plain weave (unit cell size 2.5 mm × 2.5 mm); (b) flax fiber twill weave (unit cell size 2.6 mm × 2.6 mm); (c) alumina fiber quasi-unidirectional weave (unit cell size 0.84 mm × 2.0 mm); (d) 3D woven glass weave (unit cell size 3.6 mm × 3.8 mm); (e) glass fiber knit (loop spacing 1.2 mm); (f) noncrimp carbon fiber composite (fiber bundle width 5.0 mm); (g) random steel fiber assembly (fiber diameter 8 μm); (h) carbon fiber laminate produced by automated tape placement (thickness of a ply ~0.2 mm); and (i) open glass fiber knit (loop spacing 5.5 mm).

However, after 25 years of intensive research,<sup>9,10</sup> it is now clear that CNT addition to polymers as fillers does not lead to nanocomposites that can compete with carbon fiber-reinforced polymers in mechanical performance. This is despite the exceptional stiffness and strength of CNTs. The dispersion route currently employed and other processing challenges limit achievable volume fractions and do not allow unidirectional alignment of the nanoreinforcement. Other technologies are on the rise where these difficulties are currently being addressed (see the section on CNT fibers).

In parallel with the previously discussed developments, CNTs have also been explored as a second reinforcing fiber in composites to complement existing (micro)fibers.<sup>11,12</sup> Conventional laminated composites are susceptible to damage in the form of matrix cracks and delaminations (particularly under impact). This damage is highly undesirable as it limits the composite's residual performance, including compressive strength and fatigue life. The benefit of CNTs is seen in targeting these long-standing challenges in composite performance. CNT integration can be grouped into the following three strategies: (1) deposition on fiber and ply surfaces; (2) dispersion in the matrix; and (3) integration in the fibers.

Over the years, there have been several reviews<sup>13–15</sup> and books<sup>16,17</sup> written on hierarchical composites with CNTs. We present some of the success stories and remaining challenges

in these developments, with a focus on polymer-based hierarchical composites with high stiffness, strength, and toughness enabled by CNTs.

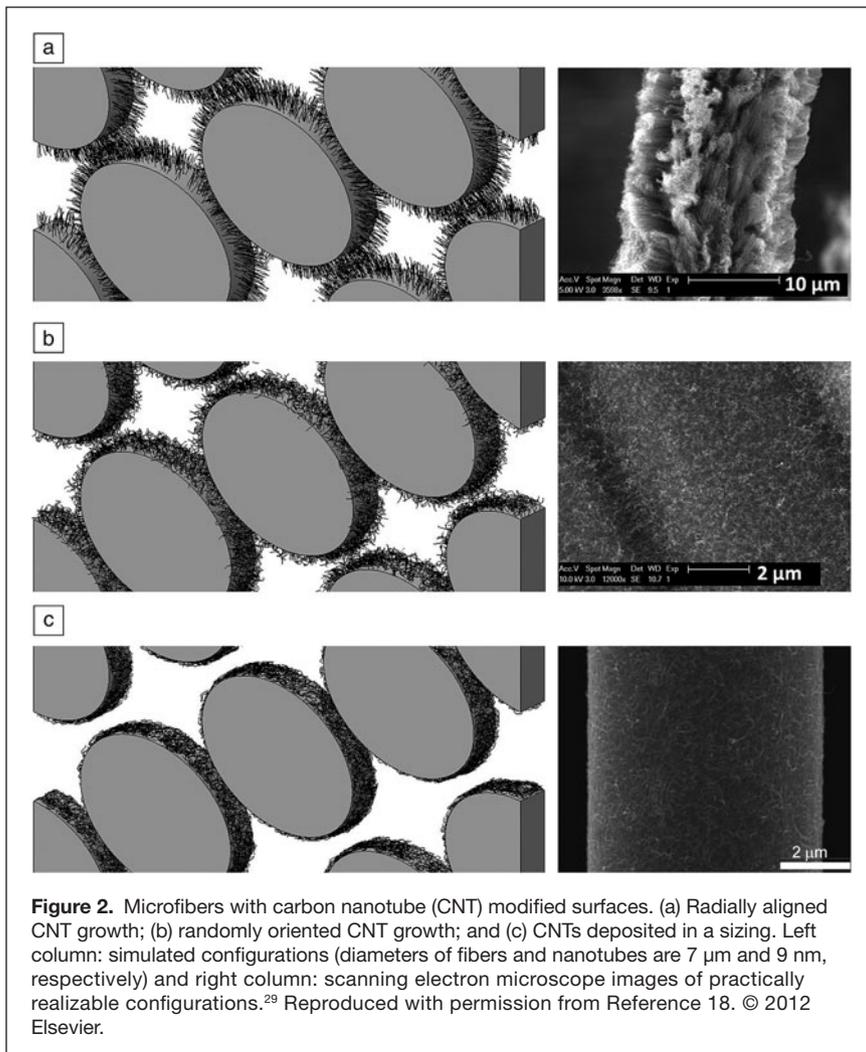
### Interface nanoengineering

Interfaces in composites are a key to engineering synergistic properties—properties that go beyond those that are brought by the two constituents (fibers and matrix). Composite properties such as compressive strength, impact resistance, fatigue life, and fracture toughness are sensitive to the performance of the fiber/matrix interface. The strength and toughness of the interface require proper tuning to achieve overall optimal performance of the composite. Fiber debonding and pullout are important energy-dissipating mechanisms that can be exploited for improving composite toughness.

The use of CNTs in composites presents many opportunities for interface engineering. In the literature, two routes are distinguished: (1) direct growth of CNTs on fibers or plies using the chemical vapor deposition (CVD) process and (2) application of already synthesized CNTs using sizing or coating technologies (**Figure 2**).<sup>18</sup> Fibers with grown CNTs are also known as “fuzzy” fibers.<sup>19</sup> The direct growth of CNTs offers advantages over other methods. With this approach, much higher CNT loadings (10–100 times higher than by

mixing) in the composite can be achieved. Also, challenges related to CNT agglomeration, filtration, and increase in matrix viscosity are common in the dispersion route, but are not present in composites with the fuzzy fibers. There are additional benefits, such as the possibility to control CNT alignment. For example, it is feasible to grow radially oriented CNTs on a fiber surface.<sup>20–22</sup> The radial alignment offers substantial benefits in terms of stiffening, strengthening, and toughening of the fiber/matrix interface,<sup>23–27</sup> which has an impact on macroscopic properties, such as interlaminar shear strength, interlaminar toughness, and composite strength. According to modeling studies, CNT forests on fibers suppress stresses at the interface,<sup>28,29</sup> allowing for a delay in damage onset.

CVD deposition of CNTs on fibers and plies is not without its challenges. One such challenge is degradation of the carbon fiber strength after the CNT growth.<sup>23–25</sup> The decrease has been reported to be as much as 50% in strength, attributed to damage on the fiber surface introduced by the catalyst particles required for the CVD process. This challenge has persisted over the years, but due to significant research efforts, it has now been addressed, and CNT growth on carbon fibers with preserved fiber performance has been successfully carried out by several research teams.<sup>30–32</sup> Producing “fuzzy” fiber composites with existing processing routes remains an open area of investigation. There is a need to better understand effects



**Figure 2.** Microfibers with carbon nanotube (CNT) modified surfaces. (a) Radially aligned CNT growth; (b) randomly oriented CNT growth; and (c) CNTs deposited in a sizing. Left column: simulated configurations (diameters of fibers and nanotubes are 7  $\mu\text{m}$  and 9 nm, respectively) and right column: scanning electron microscope images of practically realizable configurations.<sup>29</sup> Reproduced with permission from Reference 18. © 2012 Elsevier.

of the CNT fuzz on permeability, formability, and compressibility of the reinforcement, as changes in the reinforcement behavior may require adjustment of processing conditions.<sup>33–36</sup>

Localization of CNTs at interfaces can also be done via sizing or coating routes. The surface of fibers is treated (sized) for various reasons, for example, to improve their handleability or compatibility with the polymer. In this case, CNTs do not introduce damage to the fiber surface and, therefore, do not degrade the strength of the fibers. In contrast, CNT-reinforced sizings are known for their healing effects on surface defects. For glass fibers, this healing results in higher fiber strengths in comparison to fibers not sized.<sup>37,38</sup> CNT-modified sizings have also been shown to strengthen the fiber/matrix interface<sup>39,40</sup> and improve composite properties that are interface controlled. This approach is more readily scaled up from a laboratory to industrial scale.

The interlaminar interface is another critical interface in composites. It can be strengthened and toughened by positioning CNTs in between laminate layers. This can be done either via the direct CNT growth route or sizing/coating technologies. Different levels of property improvement have been

practically realized.<sup>15,41</sup> Aligned CNTs are most effective as they lead to fracture toughness improvement of up to 2–3 times.<sup>42</sup> In one study, CNTs in thermoplastic interlayer films provided up to a 30% increase in the residual laminate strength after impact.<sup>43</sup> The main mechanism claimed to be responsible for these improvements is the pulling out of CNTs upon crack propagation through the material. CNTs have a high surface area, therefore, their debonding from the matrix is an efficient source of energy dissipation. Theoretical models predict toughness improvements by two orders of magnitude in comparison with conventional carbon fibers.<sup>44</sup> Other mechanisms like crack deviation, crack pinning, and plastic deformation of the matrix have also been suggested.<sup>45,46</sup>

### Matrix nanoengineering

The toughness of fiber-reinforced composites is sensitive to the performance of the matrix. Stronger and tougher matrices lead to composites with higher resistance to matrix cracking and delaminations. Randomly oriented CNTs (implemented as matrix nanofillers) have been noted to increase, sometimes substantially, matrix-dominated properties of laminates, including the threshold for the onset of transverse cracking,<sup>47,48</sup> interlaminar fracture toughness,<sup>49,50</sup> and fatigue life.<sup>15,51–54</sup> The range of improvements varies greatly—from a few to hundreds of percent. The compressive strength after impact has been much more difficult to improve even when matrix performance

and interlaminar fracture toughness have been significantly increased.<sup>55–57</sup> Transferring improvements on the micro- and mesolevels to improved performance at the macrolevel remains a fundamental challenge of this approach.

The dispersion route of CNTs in the matrix comes with additional challenges. CNTs tend to agglomerate, which has a detrimental effect on the mechanical performance of the matrix and the composite. The issue of agglomeration becomes more severe for higher CNT loadings (or volume fractions). Therefore, practically realizable CNT concentrations are limited to a few percent by weight (sometimes even to a fraction of a percent). The final state of CNT dispersion depends on many factors, including storage history, processing steps, and curing conditions.<sup>58</sup> This makes it difficult to guarantee the same state of dispersion from one experiment to another. Impregnation of fibrous preforms with CNT modified resins is plagued by filtration of CNTs on the boundaries of fiber bundles. This is commonly related to the state of CNT dispersion.<sup>59</sup> A wet-powder impregnation is an alternative route to manufacture fiber-reinforced composites with high CNT loadings,<sup>60</sup> where the disadvantages are mitigated.

## Fiber nanoengineering

In the last five years, there has been an increasing interest in forming CNT-based fibers and sheets from well-aligned and highly packed CNTs. By infiltrating these with a polymer, nanocomposites with high mechanical performance suitable for structural applications can be produced. There are many different techniques for producing CNT-based fibers: spinning (from a CNT solution, an aligned CNT forest, or a CNT aerogel) and twisting/rolling from a CNT film.<sup>61,62</sup> Some of these techniques lead to fibers with impressive stiffness and strength, comparable to those of conventional carbon fibers. The science of making high-performance CNT-based fibers is quite complex. The fiber performance depends on properties of individual CNTs (their length, the number of walls, diameter, and waviness) and the arrangement and compaction of the CNTs inside the fiber. The internal morphology of the CNT fibers varies significantly. Once CNT fibers are made, they can be further spun into yarns<sup>63–65</sup> and processed into different preforms. Conventional textile methods like knitting, braiding, and weaving are readily available for this.

CNT fibers can be designed to achieve not only high stiffness and strength, but also high strains-to-failure, therefore their composites are expected to be much tougher than the state-of-the-art advanced composites today. CNT fibers have additional interesting properties. They are highly flexible and have a knot efficiency (the ratio of the strength of a knotted fiber to that of an unknotted fiber) as high as 100%.<sup>66</sup> The yarn-like nature of these fibers allows them to bend without permanent damage.

One of the challenges is related to the lack of common standards for mechanical characterization of the CNT fibers.<sup>61</sup> Determining the stress level in fibers with a complex morphology is not straightforward. Different approaches are used in the literature and this hinders comparison of different fibers. The relationship between processing parameters, the fiber nanostructure, and resulting macroscopic performance is not well understood and therefore is presently not well controlled. Consistent quality of individual CNTs also remains a challenge.

On the composite side, there are difficulties in achieving high-quality impregnation while avoiding dry areas and porosity. CNT fibers have low permeability and their complete wetting is not trivial.<sup>67</sup> High-pressure processes like pultrusion have been successful in providing good impregnation.<sup>68</sup> Because CNT fibers have a heterogeneous morphology, the interface with the polymer is a complex object and not well defined in the traditional sense. Implications of this for stress transfer are not clear. Once a fundamental understanding of these mechanisms is developed, the

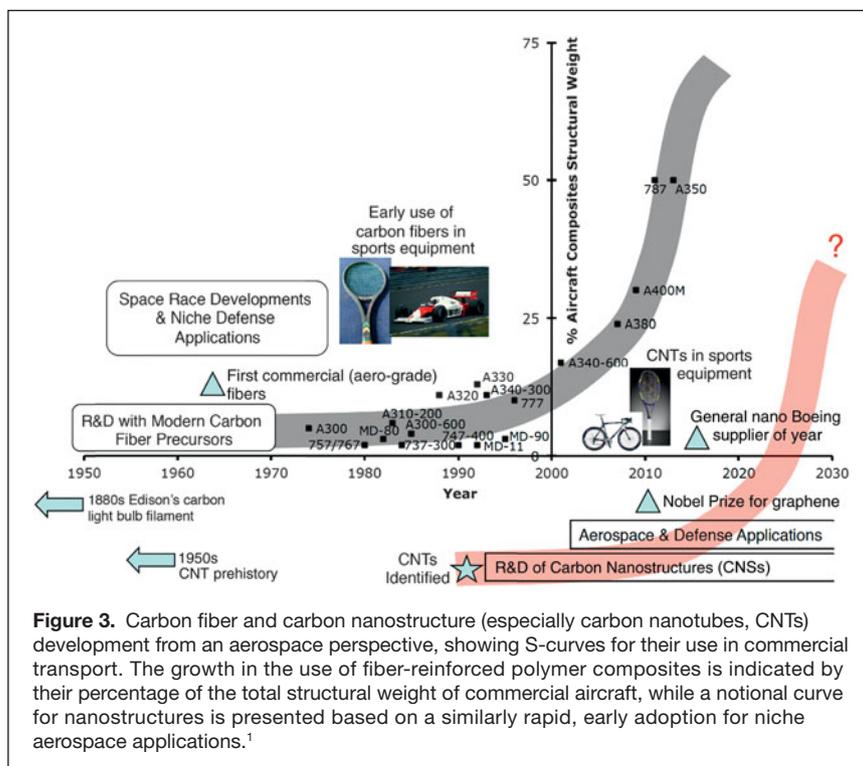
technological challenges should be overcome and the future of these novel CNT fibers looks bright.

## Summary and future outlook

Hierarchical design down to the nanoscale is a way to engineer light, stiff, strong, and super-tough structural composites. The three routes for integration of CNTs in fiber-reinforced composites (interface, matrix, fiber) have shown promising results towards achieving this goal, each with their own advantages and challenges. The most impressive improvements were achieved with aligned and highly packed CNTs.

The future of carbon nanomaterials in advanced applications is compared in **Figure 3** to the development of CFRP composites and their penetration into different markets, primarily commercial aerospace, where an S-curve is observed.<sup>69</sup> The parallels of micron-scale carbon fiber and nanometer-scale CNTs are many, both in terms of materials science and their development and adoption. Carbon fibers were considered in the early days for aerospace and military applications, many involving undisclosed or recently disclosed projects, running in parallel to lower “grade” materials and applications such as sports equipment. CNT-comprised materials and, more broadly, carbon nanostructures, are following a similar but perhaps accelerated trajectory. In this context, the rate of progress of adoption of carbon nanostructures is relatively quick providing a bright outlook for increased development. It is striking that the development of hierarchical composites has largely been experimentally driven thus far.

Modeling studies of composites with reinforcement on two length scales are scarce. The significant difference in diameters



**Figure 3.** Carbon fiber and carbon nanostructure (especially carbon nanotubes, CNTs) development from an aerospace perspective, showing S-curves for their use in commercial transport. The growth in the use of fiber-reinforced polymer composites is indicated by their percentage of the total structural weight of commercial aircraft, while a notional curve for nanostructures is presented based on a similarly rapid, early adoption for niche aerospace applications.<sup>1</sup>

of microfibers and CNTs (close to 1000 times) leads to numerical difficulties when these reinforcements need to be modeled simultaneously, but such models are now being developed.<sup>70–73</sup> They provide opportunities to investigate a large number of parameters and to simulate scenarios that are not yet possible to realize experimentally. For example, one of the models suggests that spatially resolved CNT networks are effective for the mitigation of stress concentrations at the microscale.<sup>70</sup> These findings are in line with principles that nature uses to design high-performance biological composites.<sup>74</sup> Exploiting CNT networks that are interconnected with fibers promises to yield fundamental changes in the failure process, thereby affecting properties.

The field of hierarchical composites is a young one and is taking its first steps, with many interesting developments to come. Until now, the community has focused on hierarchies utilizing only two scales of carbon materials (microscopic fibers and nanoreinforcements). Nature does it in much more intricate ways with more material combinations and more hierarchy levels. Further research is needed to understand the role of gradient nanostructures and how they are optimized for multiple properties. With the emergence of nanomaterials such as the two-dimensional graphene, it is now possible to investigate synergistic effects between different types of nanoscale reinforcements on the performance of structural composites.

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**BULK SAMPLE MEASUREMENT: TABLET**

Powder X-ray diffractometers can readily measure bulk samples if techniques are devised for sample preparation, and measurement conditions are properly set. When measuring samples which have curved surfaces, such as pharmaceutical tablets and ball bearings, various effects are seen due to the curvature, such as shifting of the diffraction angle and widening of the full width at half maximum. In the case of samples with these kinds of surfaces, adjustment is done using clay or cellophane tape from the back side of the aluminum sample plate, so that the sample surface is at the same height as the reference surface of the sample plate. The effects of curvature can be reduced by performing the measurement while narrowing the divergence slit and the incident height-limiting slit. Fig. 1 shows the X-ray diffraction pattern and qualitative analysis results for a pharmaceutical tablet. As a result of measurement, it was possible to identify the active ingredients in the tablet as ibuprofen and ethenzamide.

**APPBYTE**  
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Figure 1: X-ray diffraction patterns of pharmaceutical tablets

Apparatus condition: MiniFlex600 (FF tube 40 kV, 15mA), Detector: D/teX Ultra, Slit conditions: DS = 0.1 mm, SS = 8 mm, RS = 13 mm, Incident side and receiving side Soller slit: 2.5°, Incident height limiting slit = 2 mm  
 Measurement condition: Scan range: 2θ = 3 ~ 40°, Step width: 0.02°, Scan speed: 4° / min. (about 10 min.)